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Small contact resistance and high-frequency operation of flexible low-voltage inverted coplanar organic transistors

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The contact resistance in organic thin-film transistors (TFTs) is the limiting factor in the development of high-frequency organic TFTs. In devices fabricated in the inverted (bottom-gate) device architecture, staggered (top-contact) organic TFTs have usually shown or are predicted to show lower contact resistance than coplanar (bottom-contact) organic TFTs. However, through comparison of organic TFTs with different gate-dielectric thicknesses based on the small-molecule organic semiconductor 2,9-diphenyl-dinaphtho[2,3-*b*:2',3'-*f*] thieno[3,2-*b*]thiophene, we show the potential for bottom-contact TFTs to have lower contact resistance than top-contact TFTs, provided the gate dielectric is sufficiently thin and an interface layer such as pentafluorobenzenethiol is used to treat the surface of the source and drain contacts. We demonstrate bottom-contact TFTs fabricated on flexible plastic substrates with record-low contact resistance (29 Ω cm), record subthreshold swing (62 mV/decade), and signal-propagation delays in 11-stage unipolar ring oscillators as short as 138 ns per stage, all at operating voltages of about 3 V.

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o enable the adoption of organic thin-film transistors (TFT) in high-frequency device applications, the contact resistance must be reduced well below the smallest values reported to date¹⁻³. The reason is that the contact resistance is a key limiting factor determining the transit frequency of organic TFTs^{4,5}, and when the TFT dimensions are in the range required for megahertz operation at low voltages, the contact resistance is more limiting to the transit frequency than the intrinsic carrier mobility². For example, achieving a transit frequency of 10 MHz in a TFT with a channel length of 1 µm and a total gate-to-contact overlap of 10 µm operating with voltages of 3 V requires the contact resistance to be smaller than 40 Ω cm, regardless of whether the intrinsic channel mobility is 10, 100, or 1000 cm² V^{-1} s^{-1.2}. In organic TFTs, the contact resistance is greatly affected not only by the choice of materials, but also by the choice of TFT architecture. In the case of the inverted (bottom-gate) architectures, the staggered (top-contact; TC) configuration has typically provided smaller contact resistance than the coplanar (bottom-contact; BC) configuration, even for devices comprising the same materials and layer thicknesses^{6,7}. To date, the smallest contact resistances reported for TC and BC organic TFTs are 46.9 Ω cm¹ and 80 Ω cm⁸, respectively. These and other experimental observations are in line with most device simulations that predict that TC organic TFTs would generally outperform BC organic TFTs due to lower contact resistance^{7,9}. The smaller contact resistance of TC organic TFTs is ascribed primarily to the overlap between the contacts and a portion of the gate-induced carrier channel directly under the contacts, leading to more efficient charge injection¹⁰⁻¹². Additionally, the electrical conductivity in the contact regions may be enhanced by metal clusters penetrating into the semiconductor layer upon deposition of the contact metal¹³, by contact doping¹⁴, and by the relatively large area for charge injection between the contact metal and the gate-induced carrier channel (current crowding)^{10,15}.

In BC organic TFTs, the contact resistance is typically higher, since the gate field-assisted charge injection is weakened⁹. In addition, BC organic TFTs often exhibit a discontinuous coverage and poor thin-film morphology of the organic semiconductor layer along and across the edges of the contacts. Such a poor semiconductor morphology occurs, because the surface energy of the contact material usually differs profoundly from that of the gate dielectric⁶. Various approaches have been implemented to improve the wetting behavior of organic semiconductors on metal contacts, such as ozone exposure⁸, oxygen-plasma treatment¹⁶, and chemisorbed molecular monolayers¹⁷. Chemisorbed molecular monolayers show particular promise for the modification of both the gate dielectric and the contacts, because such modifications are area-selective, reproducible and when chosen carefully do not adversely affect subsequent processing steps^{18–22}. For metal contacts, thiol monolayers can improve the morphology of the organic semiconductor layer above the contacts and across the contact edges and can lower the injection barrier by tuning the work function of the metal contacts by a few hundred millielectronvolts²³⁻²⁷. The most utilized and effective molecule to date for improving the charge injection in p-channel organic TFTs is pentafluorobenzenethiol (PFBT)^{26,28}. The increase in the work function is caused by the large interface dipole created by the high density of fluorine atoms in the PFBT monolayer²⁹. The successful use of PFBT to improve the performance of p-channel BC organic TFTs has motivated investigations into other molecules capable of forming monolayers, particularly those with a large number of fluorine atoms to induce a work function shift beyond that obtained with PFBT²⁴. Despite these efforts to improve the contact-semiconductor interface of BC organic TFTs, their contact resistance is still largely inferior to that of the best TC organic TFTs¹.

However, recent drift-diffusion-based simulations performed by Zojer et al. predict that BC organic TFTs may exhibit lower contact resistances than otherwise equivalent TC organic TFTs^{11,30}, provided the energy barrier between the source contact and the organic semiconductor is sufficiently low and the gate dielectric is sufficiently thin. Given the importance of the contact resistance for the dynamic TFT performance^{1,2}, this is a potentially critical finding, but an experimental study to confirm the impact of the gate-dielectric thickness on the contact resistance has to our knowledge not yet been performed, although investigations into the effects of the gate-dielectric thickness on other organic-TFT-performance parameters are abundant^{31–34}.

Here, we fabricated BC and TC organic TFTs with different thicknesses of aluminum oxide passivated with an alkylphosphonic acid self-assembled monolayer (SAM) as the gate dielectric. We used gold for the source and drain contacts, PFBT to treat the contacts of the BC TFTs, and the vacuum-deposited small-molecule semiconductor 2,9-diphenyl-dinaphtho-[2,3b:2',3'-f]thieno[3,2-b]thiophene (DPh-DNTT)³⁵⁻³⁷. DPh-DNTT has previously shown low contact resistance in TC TFTs³⁸. We measured the contact resistance using the transmission line method (TLM) and found that when the gate-dielectric thickness is sufficiently small, the contact resistance is indeed smaller in the BC TFTs than in the TC TFTs, supporting the prediction by Zojer et al.³⁰. Further, we fabricated TFTs and circuits on flexible plastic substrates, utilizing an aluminum oxide/SAM hybrid gate dielectric with a thickness of 5.3 nm. For the TC TFTs, we measured a contact resistance of 56 Ω cm, very similar to the contact resistance reported by Yamamura et al. for this device architecture on a glass substrate¹. For the BC TFTs, we obtained an even smaller contact resistance of 29 Ω cm, the smallest contact resistance reported to date for organic TFTs using a nonelectrolyte gate dielectric³⁹. Furthermore, the signal-propagation delay, measured in flexible 11-stage unipolar ring oscillators at a supply voltage of 3.7 V, is 178 ns per stage for the TC TFTs and 138 ns per stage for the BC TFTs, confirming the benefit of a small contact resistance for the dynamic transistor performance. These are the shortest signal-propagation delays reported to date for organic ring oscillators at a supply voltage of less than 50 V, and they represent a significant step towards the use of organic TFTs in flexible low-power electronics applications.

Results

Organic TFTs with different gate-dielectric thicknesses. To maintain the highest possible degree of comparability between the performance of the BC and TC TFTs, multiple measures were taken to render the devices in the two architectures as equivalent as possible. This is necessary to be able to base the comparison of contact resistance on controlled assumptions. As a first measure to maintain comparability, we fabricated the TFTs in close proximity to each other on a common substrate, utilizing the same gate-oxide layer and the same semiconductor layer (Fig. 1). With this measure in place, we only directly compare those contact resistances that are extracted from TFTs fabricated on the common substrate, i.e., from TFTs sharing the same gatedielectric thicknesses. Secondly, we settled on a common nominal thickness of the vacuum-deposited DPh-DNTT (Fig. 1a) layer of 20 nm for all TFTs in this study. This is the optimum semiconductor-layer thickness that we have previously identified for TC organic TFTs based on DPh-DNTT³⁷. Note that the semiconductor-layer thickness is relevant for the contact resistance in two ways. In TC organic TFTs, a larger thickness is expected to increase the contact resistance due to the poor vertical carrier transport and the increase in trap-state density with increasing thickness of small-molecular-semiconductor



Fig. 1 Organic thin-film transistors fabricated for this study. **a** Chemical structures of pentafluorobenzenethiol (PFBT) used to treat the surface of the gold bottom contacts, of *n*-tetradecylphosphonic acid used for the gate-dielectric self-assembled monolayer (SAM), and of the organic semiconductor 2,9-diphenyl-dinaphtho-[2,3-b:2',3'-f]thieno[3,2-b]thiophene (DPh-DNTT). **b** Schematic cross-section of bottom-contact and top-contact organic TFTs fabricated on silicon substrates to study the relation between the gate-oxide thickness and the contact resistance. **c** Schematic cross-section of bottom-contact organic TFTs fabricated on PEN substrates to evaluate the contact resistance and the dynamic performance of TFTs and circuits on flexible plastic substrates



Fig. 2 Morphological characterization of vacuum-deposited DPh-DNTT films. **a** AFM topography image of a DPh-DNTT layer formed on the Al_2O_3 /SAM gate dielectric. Individual terrace levels are indicated with numbers 1-7. The scale bar corresponds to 100 nm. **b** Analysis of the height differences between the terraces. The step height of 2.6 nm corresponds closely to the length of a DPh-DNTT molecule, indicating that the molecules in each layer are oriented approximately upright with respect to the gate-dielectric surface. The data points and the error bars were obtained by local averaging of the terrace height over a large number of locations on each terrace in the AFM height image. **c** SEM image of the contact edge of a DPh-DNTT bottom-contact TFT with PFBT-treated gold contacts. The terrace morphology of the DPh-DNTT film is distinguishable on the contact surface near the contact edge (red box) and in the channel region (blue box). The scale bar corresponds to 200 nm. **d** Grazing incidence X-ray diffraction measurements on 30-nm-thick DPh-DNTT films on surfaces consisting of Au (black), PFBT-treated Au (red), and atomic-layer-deposited Al_2O_3 passivated with *n*-tetradecylphosphonic acid (PA-SAM, blue). The (110), (020), and (120) peaks are clearly distinguished in both of the latter two cases, signifying in-plane π - π stacking⁴⁸, while on bare Au only the (110) peak is present

layers^{1,12,40–42}. The intrinsic carrier mobility in the charge accumulation region, in turn, partially determines the bulk resistivity component of the contact resistance in the context of current crowding¹². The importance of this effect will depend on the particular semiconductor and the TFT architecture, since a high intrinsic mobility in the accumulation region can mitigate the larger space-charge limitations on injection in the staggered

configuration^{12,43}. As a third measure, we omitted the use of contact doping for the TC TFTs, because it would compromise the comparability to the BC organic TFTs. Contact doping has shown the potential to reduce space-charge limitations of the contact resistance in TC TFTs, possibly by reducing the width of the Schottky barrier at the contact-semiconductor interface and by filling trap states in the semiconductor region directly under



Fig. 3 Bottom-contact and top-contact DPh-DNTT TFTs with different gate-dielectric thicknesses. The TFTs have gate-oxide thicknesses (t_{ox}) of 3, 30, 50, and 100 nm, a channel length of 50 µm, and a channel width of 200 µm. **a**, **b** Transfer characteristics measured in the linear regime of operation ($V_{DS} = -0.1$ V). **c-f** Channel-width-normalized contact resistance (R_CW) of TFTs with channel lengths ranging from 6 to 50 µm extracted in the linear regime of operation ($V_{DS} = -0.1$ V) and plotted as a function of the gate-overdrive voltage ($V_{GS} - V_{th}$), showing the influence of the gate-dielectric thickness on the contact resistance. For the smallest gate-oxide thickness of 3 nm, the contact resistance of the bottom-contact (BC) TFTs is smaller than that of the top-contact (TC) TFTs

the contacts and in parts of the channel region adjacent to the contacts, with charges generated by the dopant^{20,44,45}. Finally, PFBT (Fig. 1a) was used to modify the gold bottom contacts, which primarily serves to maintain similar semiconductor morphology across the contact-to-channel interface^{26,46}. The use of PFBT has an additional benefit in terms of the charge-injection barrier at the contact-semiconductor interface of the BC TFTs²⁴.

Unfortunately, the PFBT treatment is not applicable to TC TFTs due to the buried contact-semiconductor interface, but it is nonetheless an acceptable measure for the BC TFTs in this comparative study for three reasons. First, due to the high surface energy of the gold contacts, the bottom contacts must be modified in some way to enable a fair comparison of contact resistance between these architectures at all. Second, the very large contact areas in the TFTs on the silicon substrates ensures that the largest contributor to the contact resistance in the TC TFTs is the bulk resistance under the contacts and not the interface resistance¹⁰. Finally, an Ohmic contact resistance is required to justify the use of TLM to evaluate the contact resistance⁴⁷. To that end, gold contacts were used for the TC TFTs, since the work function of gold (5.0 eV) is close to the HOMO energy level of DPh-DNTT (5.3 eV)⁴⁸, and because penetration of gold clusters into smallmolecule semiconductor films has been shown to reduce the dipole barrier that can otherwise form at the contactsemiconductor interface⁴⁹. In the BC TFTs, Ohmic contact resistance can be realized by treating the gold contacts with an interface layer, which can increase the work function. This is accomplished with PFBT, since the large dipole moment pointing towards the -SH bonding group increases the effective work function of the gold to around 5.4 eV^{24,50}.

Semiconductor thin-film morphology. Atomic force microscopy (AFM) and scanning electron microscopy (SEM) analyses of the vacuum-deposited DPh-DNTT films show that the semiconductor morphology is very similar on all substrates, regardless of the type of substrate (silicon or PEN), the method by which the gate oxide was formed (atomic-layer deposition or plasma oxidation), and the gate-oxide thickness. The DPh-DNTT films show the characteristic terrace-like structure (Fig. 2a-c) that has been observed for this and other small-molecule semiconductors^{37,38} and is indicative of in-plane π - π stacking. This is additionally confirmed by grazing incidence X-ray diffraction (GIXRD) measurements of the DPh-DNTT films deposited onto the dielectric surface (see Fig. 2d)⁴⁸. In the particular case of the BC TFTs, SEM, and GIXRD reveal that the treatment of the gold contacts with PFBT promotes the extension of the terrace-like DPh-DNTT film morphology in the channel region along and across the source and drain contact edges (Fig. 2c). Conversely, the GIXRD spectrum from DPh-DNTT deposited onto bare gold shows only the (110) peak, indicating poor in-plane π - π stacking on the gold surface.

Gate-dielectric thickness and contact resistance. Representative measured transfer characteristics of the BC and TC TFTs fabricated on silicon substrates are shown in Fig. 3a, b. Regardless of gate-dielectric thickness and the TFT architecture, all TFTs have a

threshold voltage close to zero, a subthreshold swing smaller than 200 mV/decade, and an on/off current ratio of about 10⁶. To quantify the contact resistance of the TFTs, we employed the widely used transmission line method (TLM)47. The TLM assumes that the total device resistance (R) is the sum of a channel-length-independent and Ohmic contact resistance (R_c) comprising both the source and drain contact resistances and a channel resistance proportional to the channel length $(L)^{47}$. The channel-width-normalized resistance (RW) is determined at a drain-source voltage (V_{DS}) as close to zero as possible (here: $V_{DS} = -0.1 \text{ V}$ for a set of TFTs with channel lengths ranging from 6 to 50 µm. Based on the above-mentioned assumptions, the TLM deduces, for each desired gate-overdrive voltage ($V_{GS} - V_{th}$), a channel-width-normalized contact resistance (R_CW) from the linear dependence on L of RW. In our TLM results, the linear fits of RW versus L for all gate-oxide thicknesses and gate-overdrive voltages are of good quality, with adjusted R² values > 0.9 (Supplementary Figure 1). Regardless of the device architecture and the gate-oxide thickness (t_{ox}) , R_CW is always smaller than 1 k Ω cm at sufficiently large $V_{GS} - V_{th}$, indicative of a small injection barrier.

The contact resistances obtained for the two device architectures and the four gate-oxide thicknesses are plotted in Fig. 3c-f as a function of the gate-overdrive voltage. As can bee seen, the difference between the contact resistances of the bottom-contact and top-contact TFTs depends on the gate-oxide thickness. It is intriguing to compare these results to a prediction obtained by two-dimensional drift-diffusion simulations, according to which the difference between the contact resistances of BC and TC TFTs will scale with the gate-dielectric thickness³⁰. Indeed, when the gate-oxide thickness is large (≥50 nm), the contact resistance of our TC TFTs is smaller than that of our BC TFTs over the entire range of $V_{GS}-V_{th}$. At a medium gate-oxide thickness (30 nm), the contact resistances are approximately equal. When the gate-oxide thickness is sufficiently small (3 nm), the contact resistance in the BC TFTs is significantly smaller than in the TC TFTs over the entire range of gate-overdrive voltages. This confirms that there is an opportunity for BC TFTs to outperform TC TFTs in terms of contact resistance, provided the gate dielectric is sufficiently thin and the injection barrier at the contact-semiconductor interface is small.

In addition to the observation that the difference between the contact resistances of the BC and TC TFTs depends on the gateoxide thickness, there are also noticeable differences in the shapes of the curves showing the contact resistance as a function of the gate-overdrive voltage (Fig. 3c-f): At small $V_{GS}-V_{th}$, the slope of the $R_C W$ vs. $V_{GS} - V_{th}$ curve is always steeper for the BC than for the TC TFTs, whereas at large $V_{GS}-V_{th}$, the slope is nearly the same when $t_{ox} = 100$, 50, or 30 nm (note the logarithmic scaling of $R_C W$ in Fig. 3c-f). For $t_{ox} = 3$ nm, the slope is always smaller for the BC TFTs and is also the smallest overall. We postulate that these features are related to the dependence of the contact resistance on two factors whose relative contributions depend on the TFT architecture: the geometry-specific electric-field distribution at the contact interface and the bulk resistance of the semiconductor layer between the contacts and the channel. As mentioned previously, the bulk resistance contributes significantly more strongly to the contact resistance in TC than in BC TFTs^{12,51}. In TC TFTs, the bulk resistance is primarily modulated by the thickness of the semiconductor layer, i.e., a greater semiconductor-layer thickness results in a larger access resistance. This would likely result in a larger gate-oxide thickness below which BC TFTs have lower contact resistance than otherwise comparable TC TFTs. For BC TFTs, on the other hand, the contact resistance is primarily determined by the hole-injection barrier at the contact-semiconductor interface and the presence of





Fig. 4 DPh-DNTT TFTs fabricated on flexible PEN substrates. **a** Optical microscopy image of a bottom-contact TFT with a channel length of 8 μ m, a total gate-to-contact overlap (sum of the gate-to-source and gate-to-drain overlaps) of 10 μ m, and a channel width of 200 μ m on a flexible PEN substrate. The scale bar corresponds to 100 μ m. **b** SEM image of the DPh-DNTT film in the channel region of a bottom-contact TFT on the same substrate. The terrace-like morphology of the organic semiconductor film is clearly distinguished and similar to that formed on the silicon substrates with atomic-layer-deposited aluminum oxide. The scale bar corresponds to 200 nm

space charges in the semiconductor⁴³. Therefore, the contact resistance depends strongly on the electric field and less so on the semiconductor-layer thickness^{30,52,53}. In addition, the contact resistance in BC TFTs has been shown to be dependent on the carrier mobility of the semiconductor. If the mobility is small and the charge injection efficient, space charges may build up that will inhibit carrier flow away from the region directly adjacent to the source contact, especially at small gate-source voltages^{40,43}. Hence, for the BC TFTs with $t_{ox} = 3$ nm, the contact resistance and its dependence on the gate-overdrive voltage are reduced overall.

Low-voltage bottom-contact and top-contact TFTs on flexible PEN substrates. We next show that the small contact resistance of bottom-contact DPh-DNTT TFTs with very thin gate dielectrics is evident also in TFTs fabricated on flexible plastic substrates (schematic shown in Fig. 1c). A photograph of a BC TFT fabricated on a PEN substrate is shown in Fig. 4a. An SEM image of the channel region of a DPh-DNTT TFT on a PEN substrate (Fig. 4b) indicates that the thin-film morphology of the DPh-DNTT films on the PEN substrates on which the plasma-grown aluminum oxide is used for the gate dielectric is similar to the



Fig. 5 TLM analysis of bottom-contact and top-contact DPh-DNTT TFTs fabricated on flexible PEN substrates. The TFTs have a gate-dielectric thickness of 5.3 nm, channel lengths ranging from 8 to 60 μ m, and a channel width of 200 μ m. **a**, **b** Linear fits to the total width-normalized resistance (*RW*) at selected gate-overdrive voltages ($V_{GS}-V_{th}$). **c** Width-normalized contact resistance (*R_CW*) plotted as a function of the gate-overdrive voltage. The error bars correspond to the standard error from the linear regression at each gate-overdrive voltage. **d** Intrinsic channel mobility (μ_0) plotted as a function of the gate-overdrive voltage

Table 1 Summary of t fabricated on flexible	the results of the TLM measuremen PEN substrates	ts performed on top-contact an	d bottom-contact DPh	DNTT TFTs
Device architecture	$R_{C}W$ (Ω cm) at $V_{GS}-V_{th}=-2.5 V$	L_T (μ m) at $V_{GS} - V_{th} = -2.5$ V	μ_0 (cm ² V ⁻¹ s ⁻¹)	L _{1/2} (μm)
ТС	56 ± 14	2.4 ± 0.6	5.7 ± 0.1	4.3 ± 0.2
BC	29 ± 13	1.1 ± 0.5	4.9 ± 0.1	2.6 ± 0.2
The intrinsic channel mobility (μ_{t}	$_{\rm 0}$) and the channel length at which the effective carrier	mobility is half the intrinsic channel mobility ($L_{1/}$	2) were extracted from the fits in	Supplementary Figure 2c

thin-film morphology on the silicon substrates with the atomiclayer-deposited aluminum oxide. This is to be expected, since both oxide surfaces were treated in an identical manner with an n-tetradecylphosphonic acid SAM. For the extraction of the contact resistance we performed TLM analysis on TFTs with channel lengths ranging from 8 to 60 µm, a total gate-to-contact overlap (sum of the gate-to-source and gate-to-drain overlaps, $L_{ov,total}$) of 10 µm, and a channel width of 200 µm. The results of the TLM measurements are shown in Fig. 5 and summarized in Table 1. The quality of the linear fits to the data in the RW vs. L graphs is again excellent, with adjusted $R^2 \ge 0.99$. At a gateoverdrive voltage of -2.5 V, the TC TFTs have a channel-widthnormalized contact resistance of 56 Ωcm, very similar to the record-low contact resistance reported by Yamamura et al. for TFTs fabricated in the same device architecture¹. Despite the lower intrinsic channel mobility (Fig. 5d), the BC TFTs have an

even smaller contact resistance of $29 \Omega \text{cm}$, which is to our knowledge the smallest contact resistance reported to date for organic transistors fabricated in the coplanar device architecture. For comparison, Stadlober et al. reported a contact resistance of $80 \Omega \text{cm}$ for bottom-gate, bottom-contact pentacene TFTs in which the surface of the gold source and drain contacts had been exposed to ultraviolet radiation and ozone in order to induce a favorable pentacene morphology on the contacts⁸. Braga et al. reported the smallest contact resistance yet reported for organic TFTs of any architecture in electrolyte-gated top-gate staggered poly(3-hexylthiophene) TFTs in which strong doping of the semiconductor from the electrolyte resulted in a very small contact resistance of $1 \Omega \text{cm}^{39}$.

In addition to the contact resistance, the TLM analysis also yields the transfer length (L_T) . In staggered TFTs, L_T is the contact length over which 63% of the charge-carrier exchange



Fig. 6 Static electrical characteristics of bottom-contact and top-contact DPh-DNTT TFTs on flexible PEN substrates. The TFTs have a gate-dielectric thickness of 5.3 nm, a channel length of 8 μ m, a channel width of 200 μ m, and a total gate-to-contact overlap of 4 μ m (**a**-**c**) or 10 μ m (**d**-**f**). **a**, **d** Transfer characteristics measured in the saturation regime ($V_{DS} = -3$ V). **b**, **e** Effective carrier mobility extracted from the transfer characteristics in the saturation regime ($V_{DS} = -3$ V). **c**, **f** Output characteristics of the same TFTs

μ μ _{eff,sat} (V ⁻¹ s ⁻¹	cm² SS (mV/) dec)	On/off ratio
2.0		
3.9	94	10 ⁹
4.2	92	10 ⁹
4.6	62-64	10 ⁹
4.4	68	10 ⁸
	4.2 4.6 4.4	4.6 62-64 4.4 68

occurs between the contact and the semiconductor^{12,54}. For our TC TFTs, the transfer length is 2.4 µm, which is significantly smaller than the gate-to-source and gate-to-drain overlaps ($L_{ov,S}$ and $L_{ov,D}$), signifying that the injection is not limited by the contact area⁵⁵. Despite the fact that the physical meaning of the transfer length in BC TFTs has so far not been elucidated, we still report it in Table 1 for comparison. The effective carrier mobility (μ_{eff}) extracted from the transfer curves in the linear regime of operation ($V_{DS} = -0.1$ V) is less affected by the contact resistance in TFTs with long channel lengths (Supplementary Figure 2c). The intrinsic channel mobility (μ_{0}) and the channel length at which the effective carrier mobility is half the intrinsic channel mobility ($L_{1/2}$) were extracted using Equation 1 from ref. ³⁷.

The transfer and output characteristics of DPh-DNTT TFTs with a channel length of 8 μ m and total gate-to-contact overlaps ($L_{ov,total} = L_{ov,S} + L_{ov,D}$) of 4 and 10 μ m are shown in Fig. 6. All TFTs have threshold voltages of about -1 V and on/off current ratios, here defined as the ratio between the drain current at V_{GS} = -3V and the lowest drain current measured in the transfer curve, between 10⁸ and 10⁹. The smaller contact resistance of the BC TFTs results in a larger effective carrier mobility compared to the TC TFTs (see Fig. 6 and Table 2).

All TFTs have subthreshold swings (SS) smaller than 100 mV/ decade, but those of the BC TFTs are notably smaller (62-68 mV/ decade) than those of the TC TFTs (92-94 mV/decade). For the BC TFT with the smallest subthreshold swing, we have extracted the exact subthreshold swing using two different methods: once by fitting an exponential function to the data over a range of 200 mV in the subthreshold regime and once by point-wise derivation of the measured transfer curves (Supplementary Figure 3). Depending on the method and the applied drain-source voltage, the subthreshold swing is between 62 and 64 mV/decade. To our knowledge, this is the smallest subthreshold swing reported to date for organic TFTs, regardless of device architecture, gate dielectric, and semiconductor⁵⁶⁻⁵⁹. The observation that the BC TFTs have a notably smaller subthreshold swing than the TC TFTs suggests that the subthreshold swing is affected not only by the charge-trap density at the interface between the gate dielectric and the semiconductor layer (which is nominally identical in the two device architectures), but also by the charge-trap density in the semiconductor volume that separates the top contacts from the gate-induced carrier channel which the carriers have to traverse in the TC TFTs.

Flexible low-voltage bottom-contact TFTs with small channel lengths. Often when the channel length is reduced, short-channel effects, such as drain-induced barrier lowering and increased off-state drain current, can become more prominent⁵⁵. To investigate whether our flexible BC TFTs show any of these effects, we have fabricated short-channel bottom-contact DPh-DNTT TFTs on PEN substrates. These TFTs have a channel length ranging from 0.5 µm to 10 µm, a channel width of 50 µm, and a total gate-to-contact overlap of 10 µm. Regardless of the channel length, all TFTs have an on/off current ratio of at least 10^8 (Fig. 7a), and the output curves do not show any noticeable non-linearity at small drain-source voltages that would indicate Schottky contacts (Fig. 7b)⁶⁰. All TFTs with a channel length of at least 0.8 µm show effective carrier mobilities above $1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (Fig. 7c). The



Fig. 7 Short-channel bottom-contact DPh-DNTT TFTs on flexible PEN substrates. The TFTs have a gate-dielectric thickness of 5.3 nm, channel lengths ranging from 0.5 to 10 μ m, a channel width of 50 μ m, and a total gate-to-contact overlap of 10 μ m. **a** Transfer characteristics measured in the linear regime of operation ($V_{DS} = -0.1$ V). **b** Output characteristics of the same TFTs measured at a gate-source voltage of -3 V. **c** Effective carrier mobility (μ_{eff}) plotted as a function of the channel length. The error bars correspond to the standard error from the linear regression at each gate-overdrive voltage. **d** Channel-width-normalized contact resistance (R_cW) plotted as a function of the gate-overdrive voltage ($V_{GS} - V_{th}$)

TLM analysis again shows a low contact resistance of 38 Ω cm at a gate-overdrive voltage of -2.5 V (Fig. 7d, Supplementary Figure 4).

Dynamic performance of flexible bottom-contact and topcontact TFTs. Finally, to demonstrate the benefit of a small contact resistance for the dynamic TFT performance, 11-stage unipolar ring oscillators were fabricated on the same PEN substrates as the TFTs discussed above (Fig. 8). All TFTs in the ring oscillators have a channel length of 1 µm and a total gate-tocontact overlap of 10 µm. For this channel length and gate-tocontact overlap, the effective carrier mobilities are 1.3 cm² V^{-1} s⁻¹ for the TC TFTs and 1.7 cm² V⁻¹ s⁻¹ for the BC TFTs (Supplementary Figure 5). The ring oscillators utilize the biasedload inverter design⁶¹. The signal-propagation delay (τ) is calculated from the oscillation frequency (f_{osc}) and the number of stages (n) by $\tau = 1/(2nf_{osc})^{62}$. At a supply voltage of 3.7 V, the measured signal-propagation delay is 178 ns per stage for the ring oscillator based on the TC TFTs and 138 ns per stage for the ring oscillator based on the BC TFTs, confirming the effect of the contact resistance on the frequency behavior of the TFTs. These signal delays are to our knowledge the shortest delays reported to date for organic ring oscillators on flexible substrates⁶³ and the shortest delays for organic ring oscillators on any substrate at a supply voltage of less than 50 V^{64} .

Discussion

Through an analysis of the contact resistance in TFTs with different gate-dielectric thicknesses, we have found strong experimental indications that it is possible to fabricate bottomgate, bottom-contact organic TFTs that show lower contact resistance than comparable top-contact TFTs. As predicted by Zojer et al.³⁰, we have found that for sufficiently small gatedielectric thickness, bottom-contact TFTs have lower contact resistance than top-contact TFTs, so long as sufficient measures are taken to control the semiconductor thin-film morphology across the contact-channel interface and to minimize the barrier height at the contact-semiconductor interface. This was accomplished here by employing a thin hybrid gate dielectric composed of aluminum oxide passivated with an alkylphosphonic acid SAM in combination with PFBT-modified gold source and drain contacts in the bottom-contact TFTs. The potential of this approach to improve the static and dynamic performance of organic TFTs is most significantly exemplified here by bottomcontact DPh-DNTT TFTs with a gate-dielectric thickness of 5.3 nm fabricated on flexible PEN substrates which show a channelwidth-normalized contact resistance as small as 29 Ωcm. In addition to a low contact resistance, bottom-contact TFTs can show improvements in other performance metrics, including subthreshold swings as small as 62-64 mV/decade and on/off current ratios as high as 109. Furthermore, the lower contact resistance of the bottom-contact TFTs enables higher frequencies in flexible organic-TFT circuits operating at low voltages, as shown here by the signal-propagation delay of 138 ns per stage at a supply voltage of 3.7 V, obtained in 11-stage unipolar ring oscillators based on bottom-contact DPh-DNTT TFTs fabricated on flexible PEN substrates. It is possible that even lower contact resistance is achievable with other combinations of interface



Fig. 8 Dynamic performance of unipolar ring oscillators on flexible PEN substrates. **a** Photograph of an 11-stage unipolar ring oscillator based on bottomcontact DPh-DNTT TFTs with a channel length of 1 μ m and a total gate-to-contact overlap of 10 μ m. **b** Circuit diagram of the biased-load inverters implemented for the ring oscillators. **c** Signal-propagation delay per stage measured on ring oscillators based on top-contact(TC) and bottom-contact(BC) TFTs plotted as a function of the supply voltage. **d** Output signals measured at a supply voltage of 3.7 V, showing stage delays of 178 ns for the ring oscillator based on TC TFTs and 138 ns for the ring oscillator based on BC TFTs. Depending on the supply voltage, the bias voltage is between -1and -2.5 V

layers, gate dielectrics and semiconductors in the bottom-gate, bottom-contact architecture. Further reductions in contact resistance, ideally in combination with smaller lateral TFT dimensions, are then expected to yield even higher dynamic TFT performance².

Methods

TFTs with different gate-oxide thicknesses on Si substrates. The TFTs that were used to study the relation between the gate-dielectric thickness and the contact resistance were fabricated on heavily doped silicon wafers (525 µm thickness). To reduce the effects of substrate-to-substrate variations, bottom-contact (BC) and top-contact (TC) TFTs with a common gate-dielectric thickness were fabricated on the same substrate in close proximity to each other (separated by about 100-200 µm). The silicon substrate serves as a global gate electrode for all TFTs on the substrate (Fig. 1a). As the first component of the gate dielectric, aluminum oxide (Al₂O₃) was deposited by atomic-layer deposition (ALD, Savannah 100, Cambridge NanoTech Inc.; substrate temperature 250 °C, 10 cycles/nm) with a thickness of 3, 30, 50, or 100 nm. The silicon wafers were then cut into strips $(0.5\times3~\text{cm}).$ The Al_2O_3 surface was activated by oxygen plasma (Oxford Instruments; oxygen flow rate 30 sccm, partial pressure 10 mTorr, plasma power 200 W, duration 30 s) and then passivated with a self-assembled monolayer (SAM) by immersing the substrate into a 1-mM solution of n-tetradecylphosphonic acid (PCI Synthesis, Newburyport, MA, U.S.A.) in 2-propanol (VLSI grade) for one to two hours³³. Afterwards, the substrates were rinsed in 2-propanol and dried on a hotplate (150 °C, 1 min). The capacitance of these dielectrics was calculated assuming relative dielectric constants (ϵ_r) of 9 for Al₂O₃ and 2.5 for the phosphonic acid SAM³³. Next, gold bottom source and drain contacts were deposited by thermal evaporation in vacuum onto the surface of the Al2O3/SAM gate dielectric and modified with a monolayer of pentafluorobenzenethiol (PFBT, Santa Cruz Biotechnology, Heidelberg, Germany) by immersing the substrates into a 10-mM solution of PFBT in 2-propanol for 30 min. The substrates were then rinsed with 2propanol. A 20-nm-thick layer of DPh-DNTT (Nippon Kayaku, kindly provided by Koichi Ikeda) was then deposited by sublimation in vacuum (base pressure 10⁻⁶ mbar, substrate temperature 90 °C, deposition rate 0.3 Å s⁻¹) onto all four substrates simultaneously (in order to minimize substrate-to-substrate variations). Prior to electrical measurements, the unpatterned DPh-DNTT layer was scratched away around each set of source and drain contacts using a probe needle. Electrical

measurements were then performed on the BC TFTs, followed by depositing the gold top source and drain contacts onto the organic semiconductor layer in close proximity to the bottom contacts and performing the electrical measurements on the TC TFTs. All contacts have an area of $200 \times 200 \,\mu\text{m}$ and were patterned using a silicon stencil mask (IMS Chips, Stuttgart, Germany) with channel lengths ranging from 4 to $50 \,\mu\text{m}^{38}$.

TFTs and ring oscillators on flexible PEN substrates. We fabricated bottomcontact and top-contact DPh-DNTT TFTs and 11-stage unipolar ring oscillators on 125-µm-thick flexible polyethylene naphthalate substrates (Teonex® Q65 PEN; provided by William A. MacDonald, DuPont Teijin Films, Wilton, U.K.) using a set of four silicon stencil masks (IMS Chips, Stuttgart, Germany) to define interconnects, gate electrodes, source and drain contacts, and the organic semiconductor layer (Fig. 1b) 62,65 . Prior to fabrication, the PEN substrates were baked at a temperature of 100 °C for 1 h and cleaned with 2-propanol. In the first fabrication step, 30-nm-thick gold interconnects and probe pads were deposited through the first stencil mask. For the gate electrodes, a 30-nm-thick layer of aluminum was deposited through the second stencil mask. In the case of the TFTs discussed above that were used to evaluate the relation between the gate-dielectric thickness and the contact resistance, the gate oxide was deposited by atomic-layer deposition. ALD has the advantage that the oxide thickness can be easily scaled over a wide range. However, one issue with ALD is that the oxide is not easily deposited selectively, so that subtractive patterning is usually required to create access to the gate electrodes underneath the oxide. For the TFTs discussed above, this issue was avoided by contacting the gate electrode (i.e., the doped silicon substrate) from the backside of the substrate. For devices and circuits on plastic substrates this is not an option. Therefore, for the TFTs and ring oscillators on PEN, we used plasma oxidation (Oxford Instruments, 30 sccm oxygen, 10 mTorr, 200 W, 30 s) to form a thin aluminum oxide (AlOx) layer selectively on the patterned aluminum gate electrodes. The completed gate dielectric is a stack of the 3.6nm-thick layer of AlO_x and an *n*-tetradecylphosphonic acid SAM, resulting in a total dielectric thickness of 5.3 nm and a unit-area capacitance of 0.7 µF cm^{-2.37}. For all subsequent layers, the fabrication procedure was the same as described above for the TFTs on silicon substrates, with the exception that the semiconductor layer is patterned with a stencil mask. In the ring oscillators, the drive TFTs have channel widths of 80 um and the bias TFTs have channel widths of 20 um. In the two buffer inverters prior to the output node the drive TFT has a channel width of 220 µm and the bias TFT has a channel width of 60 µm.

Semiconductor thin-film morphology characterization. The thin-film morphology of the DPh-DNTT layer was characterized using tapping-mode atomic force microscopy (AFM, Bruker Dimension Icon), scanning electron microscopy (SEM, Zeiss Merlin), and grazing incidence X-ray diffraction (GIXRD, Rigaku SmartLab). The X-ray diffractometer is equipped with a 9 kW copper source. The grazing incidence angle was set to 0.15°. The detector moved horizontally at $2\theta = 0.15^\circ$ and $2\theta\chi$ from 15° to 30° in steps of 0.1°. The X-ray beam size was set to 5×0.1 mm. The diffraction peaks were assigned using the PDXL software with the standard Gaussian distribution method. SEM and AFM were performed on completely processed TFTs, while GIXRD required samples either without any metal or with the gold layer covering the entire substrate, due to the large spot size required for the measurement. Three silicon substrates were thus prepared for the GIXRD measurements, with 30-nm thick DPh-DNTT deposited onto 30-nm thick Au, 30-nm thick Au treated with PFBT, and atomic-layer-deposited Al₂O₃ passivated with an *n*-tetradecylphosphonic acid SAM.

Electrical measurements. All electrical measurements, including the bias-stress measurements summarized in Supplementary Figure 6, were performed in ambient air at room temperature (292 K). The capacitance of the gate dielectrics, the current-voltage characteristics of the TFTs, and the signal-propagation delays of the ring oscillators were measured using an Agilent 4156 C Semiconductor Parameter Analyzer, a Tektronix TDS1000 oscilloscope, a Femto DPLCA-200 low-noise transimpedance amplifier, and gold-plated tungsten probe tips (EPP GmbH) with a tip radius of 50 μ m for contacting the probe pads.

Data availability

The data that support the findings of this study are available from the corresponding author on reasonable request.

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Author contributions

J.W.B. and H.K. devised the experimental details of the study. J.W.B. fabricated all devices and samples and performed electrical and morphological measurements. J.W.B., H.K. and K.Z. wrote the article. B.P. and P.K.L.C. performed grazing incidence X-ray characterization. F.L. and J.N.B. manufactured the silicon stencil masks used to fabricate the TFTs and circuits. S.L. and H.K. supervised the project. All the authors discussed the results and contributed to the development of the final manuscript.

Additional information

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Supplementary Information

Small contact resistance and high-frequency operation of flexible low-voltage inverted coplanar organic transistors

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Supplementary Figures



Figure 1: Analysis of DPh-DNTT TFTs with different gate-dielectric thicknesses. The bottom contact (BC) and top contact (TF) TFTs are fabricated on highly-doped cilicon substrates and have SAM-modified atomic-layer-deposited Al₂O₃ gate dielectrics with gate-oxide thicknesses of 3, 30, 50 and 100 nm, channel lengths ranging from 4 to 50 μ m, and a channel width of 200 μ m. (**a**,**c**) Transmission line method (TLM) analysis performed at the largest gate overdrive voltage for each gate-oxide thickness. All fits show R² > 0.9. (**b**,**d**) Effective carrier mobility plotted as a function of the channel length.



Figure 2: DPh-DNTT TFTs fabricated on flexible PEN substrates. The TFTs have channel lengths ranging from 8 to 60 µm, a channel width of 200 µm, and a total gate-to-contact overlap of 10 µm. (a) Transfer characteristics of the bottom-contact TFTs. (b) Transfer characteristics of the top-contact TFTs. The transfer data from **a** and **b** was employed for the TLM analysis reported in **Figure 5** and **Table 1** of the main manuscript. (c) Effective carrier mobility (μ_{eff}) plotted as a function of the channel length. The data are fit to the equation $\mu_{eff} = \mu_0/(1+L_{1/2}/L)$, where μ_0 is the intrinsic channel mobility, L is the channel length at which $\mu_{eff} = \frac{1}{2} \mu_0$.



Figure 3: Subthreshold characteristics of bottom-contact DPh-DNTT TFTs fabricated on flexible PEN. The TFTs have a channel length of 4 μ m, a channel width of 200 μ m, and a total gate-to-contact overlap of 4 μ m. (a) Exponential fit to the subthreshold region of the transfer characteristics of the TFT with the steepest subthreshold swing, yielding subthreshold swings of 62 and 64 mV/decade at drain-source voltages of -0.1 and -3 V, respectively. (b) Derivative of the measured transfer curves plotted as a function of the gate-source voltage. The dotted line denotes the theoretical limit of the subthreshold swing at a temperature of 292 K, i.e., 58 mV/decade.



Figure 4: Bottom-contact DPh-DNTT TFTs with short channel lengths fabricated on a PEN substrate. (a) Measured transfer curves of the short-channel bottom-contact DPh-DNTT TFTs fabricated on flexible PEN substrates employed for the TLM analysis reported in **Figure 7** of the main manuscript. The TFTs have channel lengths ranging from 0.5 to 10 μ m, a channel width of 50 μ m, and a total contact overlap length of 10 μ m. (b) TLM analysis performed at the largest gate overdrive voltage.



Figure5: DPh-DNTT TFTs with a channel length of 1 μ m fabricated on PEN substrates. (a,c) Transfer and (b,d) output characteristics of bottom-contact and top-contact DPh-DNTT TFTs fabricated on flexible polyethylene naphthalate (PEN) substrates having a channel length of 1 μ m, a channel width of 50 μ m, and a total gate overlap length of 10 μ m.



Figure 6: Bias-stress stability of DPh-DNTT TFTs fabricated on a silicon substrate. These TFTs were fabricated by depositing a 30-nm-thick aluminum gate electrode onto the doped silicon substrate and then forming a 3.6-nm-thick layer of aluminum oxide by oxygen-plasma growth and a 1.7-nm-thick *n*-tetradecylphosphonic acid self-assembled monolayer from solution to obtain a 5.4-nm-thick AlO_x/SAM gate dielectric. The TFTs have a channel length of 20 μ m and a channel width of 200 μ m. The bias stress was performed by applying gate-source and drain-source voltages of -2 V continuously for a duration of 64 hours in ambient air. The effective carrier mobility of the bottom-contact TFT (**a**) was 3.5 cm²/Vs prior to and 3.0 cm²/Vs after the bias-stress experiment. The effective mobility of the top-contact TFT (**b**) was 3.7 cm²/Vs prior to and 3.5 cm²/Vs after the bias-stress experiment.